

## A novel and simple approach for the synthesis of Fe<sub>3</sub>O<sub>4</sub>-graphene composite

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(Received 2 January 2012 • accepted 7 March 2012)

**Abstract**—High crystalline Fe<sub>3</sub>O<sub>4</sub>-graphene composite has been successfully synthesized *via* one-step thermolysis reaction. The results demonstrated that the attachment of iron-organic complex with graphene oxide sheets can facilely lead to magnetic graphene composites with a time-dependent calcination process. The composite was characterized by scanning electron microscopy, transmission electron microscopy, Raman spectroscopy, X-ray diffraction, and X-ray photoelectron spectroscopy. The high efficient removal of methylene blue determined that Fe<sub>3</sub>O<sub>4</sub>-graphene is a promising sorbent material for wastewater treatment.

Key words: Graphene, Fe<sub>3</sub>O<sub>4</sub>, Thermolysis Reaction, Removal

### INTRODUCTION

Magnetic graphene composites are generally obtained by graphene decorated with magnetic nanoparticles, including Fe [1], FeOH [2],  $\gamma$ -Fe<sub>2</sub>O<sub>3</sub> [3] and Fe<sub>3</sub>O<sub>4</sub> [4,5] and so on. Among them, Fe<sub>3</sub>O<sub>4</sub>/graphene composite has long been attractive because it has potential applications in enhanced optical limiting, MRI, drug delivery, Li ion batteries and removal of contaminants from waste water [4,6-9]. Since Chandra et al. reported nearly complete (over 99.9%) arsenic removal using a magnetic graphene in 2010 [10], graphene and its composites have become more and more popular in wastewater purification [11-13]. In particular, the magnetic graphene has absolute advantage for its separation from water. Although the new type of magnetic graphene has shown promising use in treating waste water, its complicated producing procedures and milligram-production-scale still limit its application in backward countries. Hence, it is still a major challenge to develop simple and facile approaches with low costs for the synthesis of magnetic graphene.

However, these complicated synthesis procedures for high crystalline Fe<sub>3</sub>O<sub>4</sub>/graphene make their impossibility in industrial production and application. Herein, we have synthesized the composite of graphene decorated with high crystalline Fe<sub>3</sub>O<sub>4</sub> nanoparticles with a novel and simple thermolysis reaction. Furthermore, taking MB removal as a model, we demonstrated that our Fe<sub>3</sub>O<sub>4</sub>-graphene composites have really great potential in wastewater treatment.

### EXPERIMENT

#### 1. Synthesis of Fe<sub>3</sub>O<sub>4</sub>-graphene Composite

Graphene oxide (GO) was prepared from natural graphite (KS-6) according to Hummer's method [14]. Fe<sub>3</sub>O<sub>4</sub>-graphene compos-

ite was produced as follows: one gram of the graphene oxide powder was dispersed into 400 mL water-alcohol solution with 1 : 1 volume ratio to yield a homogeneous brown suspension. 3.4 g bis (cyclopentadienyl) iron was then added into the graphene oxide suspension and stirred for 30 min. After physical adsorption, the complex was filtered and dried at 50 °C, and then calcined at 600 °C under an oxygen-free condition for 1 h. After being cooled to room temperature, the Fe<sub>3</sub>O<sub>4</sub>-graphene composite was obtained.

#### 2. Characterization

XRD patterns were obtained with a diffractometer (Shimada XD-D1, Japan) using Cu K $\alpha$  radiation. Raman spectra were carried out by a Horiba Jobin Yvon LabRAM using a 100 $\times$  objective lens with a 532 nm laser excitation. The state of the dispersed Fe<sub>3</sub>O<sub>4</sub> on graphene-nanosheets was observed using transmission electron microscopy (TEM, JEOL, JEM-2010, Japan). The clean transparent solution was analyzed by using a UV/Vis spectrophotometer. The spectra for each sample were recorded and the absorbance was determined at characteristic wavelength 660 nm for the each MB solution degraded.

#### 3. Adsorption Ability

5 mg Fe<sub>3</sub>O<sub>4</sub>-graphene composites were added into 20 mL 0.1 gL<sup>-1</sup> MB aqueous solution. After 20 s magnetic separation by an external magnetic field under natural lighting, the clean transparent solution was analyzed by a UV/Vis spectrophotometer, and the absorbance at characteristic wavelength 660 nm was determined for each MB solution.

### RESULTS AND DISCUSSION

XRD is a rapid analytical technique primarily used for phase identification of crystalline materials and can provide information on unit cell dimensions. Fig. 1 shows the XRD patterns of GO, graphene, Fe<sub>3</sub>O<sub>4</sub>-graphene calcined for 10 min and 60 min, respectively. It is obvious that the magnetic graphene composite formation is a time-dependent synthesis process. Since GO prepared from graphite by

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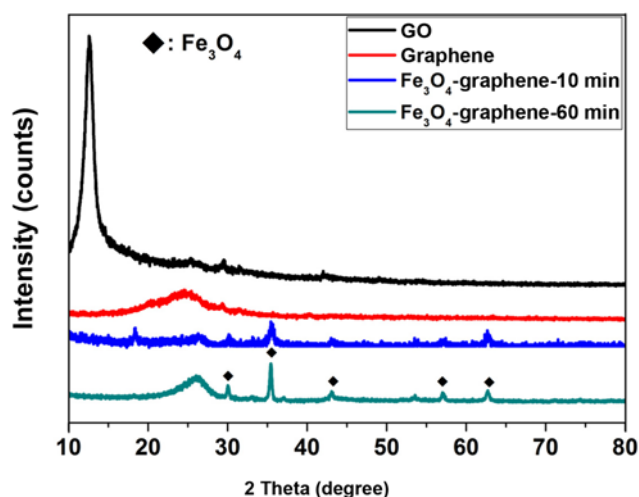


Fig. 1. XRD patterns of GO, graphene,  $\text{Fe}_3\text{O}_4$ -graphene calcined for 10 min and 60 min.

liquid-phase oxidation is one of the intercalation compounds, it can be found that the  $d$ -spacing of GO layers (0.42 nm) is larger than that of the layers of graphite (0.34 nm), as has been mentioned by Lee et al. [15]. At calcining time of 10 min, converting iron complex attached to small magnetite subparticles with weak crystallization is obtained, corresponding to the low-intensity diffraction peaks of magnetite in the XRD pattern, and the crystalline substances with

obvious diffraction peaks correspond to GO and graphene at  $18.4^\circ$  and  $26.3^\circ$ , respectively [10]. With the calcining time up to 60 min, these unstable subparticles with high surface energy tend to aggregate and grow into high crystalline magnetite nanoparticles by van der Waals forces and magnetic dipole [4], while a rising peak can index to graphene phase which is further shifted into high  $2\theta$  degree compared to pristine graphene. Because GO has many defects or nanoholes, it is reasonable that thermally reduced graphene would also have many defects and nanoholes ( $d_{\text{Df}}$ ). Therefore, GO and graphene can have oxide groups of C-Ox with some  $\text{sp}^3$  bonds in their defects and nanoholes ( $d_{\text{Ox}}$ ). GO has the largest interlayer distance ( $d_{\text{GO}}$ ) because of its intercalated  $\text{H}_2\text{O}$  molecules and various oxide groups. If defects in the layer are sufficiently large, oxide groups and  $\text{H}_2\text{O}$  molecules can be considered to exist in the empty space. On the basis of a model with various interlayers, it can be inferred that GO has an intermediate structure with  $d_{\text{Ox}}$  and  $d_{\text{Df}}$  during thermal reduction, and through bottom-up layer stacking, the resultant graphene evolves toward graphite as crystal growth with the removal of  $d_{\text{Ox}}$  and  $d_{\text{Df}}$  [16]. However, the prior formation of iron-organic complex onto GO surface by chemical interaction will reduce the removal of  $d_{\text{Ox}}$  and  $d_{\text{Df}}$  so that the XRD pattern of  $\text{Fe}_3\text{O}_4$ -graphene displays a rising peak at high  $2\theta$  degree compared to pristine graphene.

The time-dependent crystal growth was further investigated by TEM. As shown in Fig. 2(a), the  $\text{Fe}_3\text{O}_4$ -graphene composite after calcination for 10 min shows that the small magnetite subparticles were attached onto graphene surface. With calcining up to 60 min, these subparticles gradually grew to high crystalline  $\text{Fe}_3\text{O}_4$  nano-

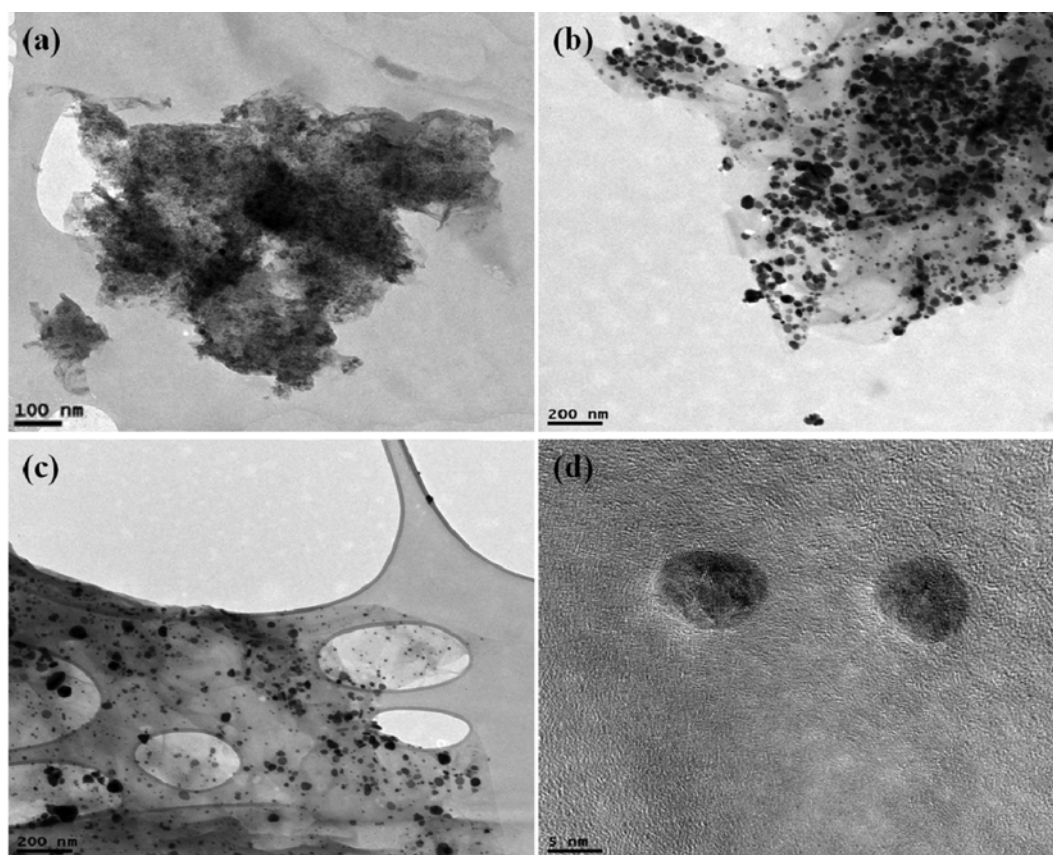


Fig. 2. TEM images of  $\text{Fe}_3\text{O}_4$ -graphene calcined for 10 min (a) and 60 min (b)-(c). HR-TEM image of  $\text{Fe}_3\text{O}_4$ -graphene calcined for 60 min (d).

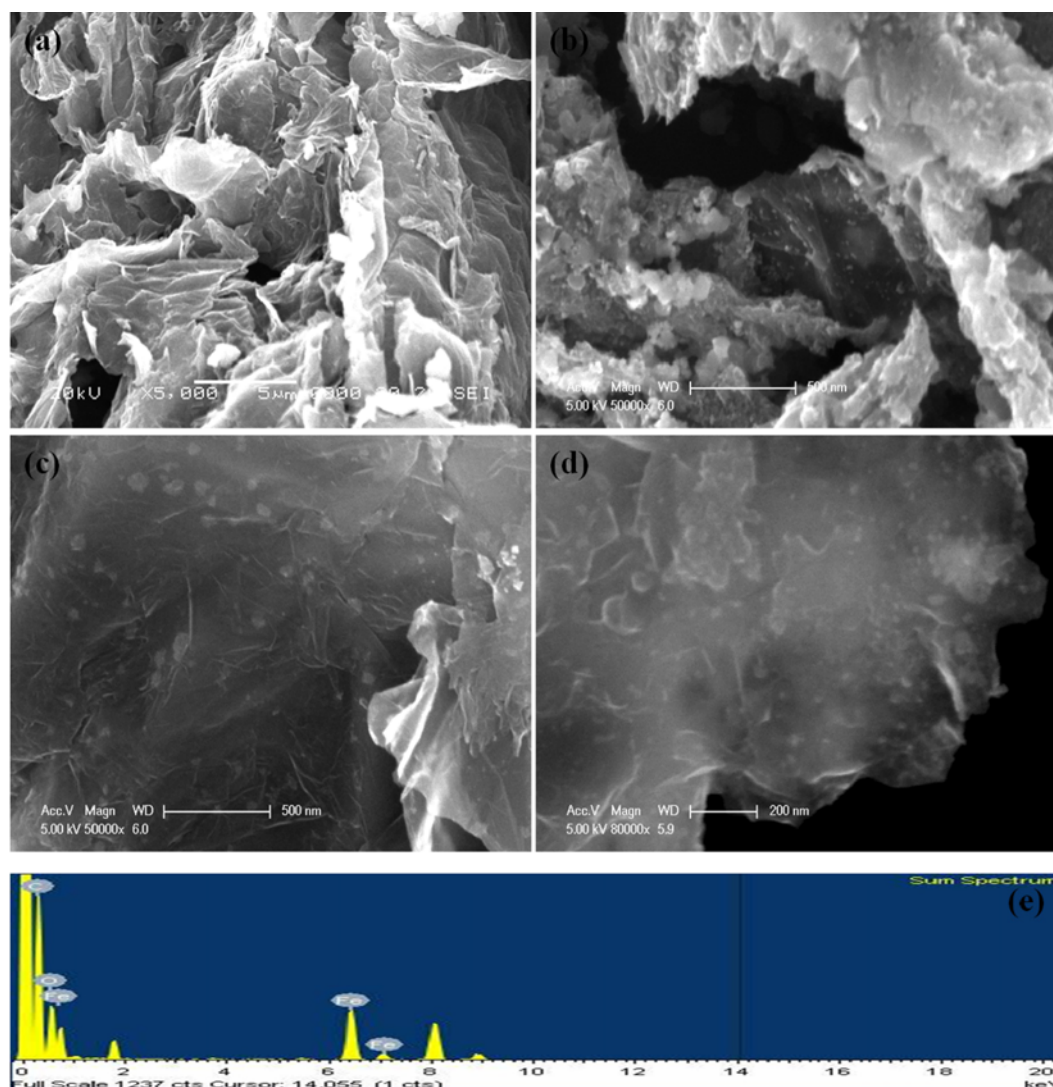


Fig. 3. SEM images of (c) graphene, and (b)-(d)  $\text{Fe}_3\text{O}_4$ -graphene. (e) EDX spectrum of  $\text{Fe}_3\text{O}_4$ -graphene.

particles by high temperature thermolysis (Fig. 2(b)-(c)). The size of the  $\text{Fe}_3\text{O}_4$  nanoparticles was found to be below 10 nm, as measured in Fig. 2(d). The microstructure of the  $\text{Fe}_3\text{O}_4$ -graphene composite was characterized by SEM. In the case of graphene, because of oxidation and heat treatment, it broke into pieces smaller than pristine graphite, as observed in Fig. 3(a). The SEM images taken from graphene and  $\text{Fe}_3\text{O}_4$ -graphene composites (Fig. 2(b)-(d)) showed that they possessed a layer-by-layer assembled structure consisting of graphene nanosheets and  $\text{Fe}_3\text{O}_4$  particles. These results could be further confirmed by EDX spectrum (Fig. 3(e)).

Fig. 4(a) shows the Raman spectra of GO and  $\text{Fe}_3\text{O}_4$ -graphene. Peaks of GO at  $1,592\text{ cm}^{-1}$  and  $1,350\text{ cm}^{-1}$  correspond to  $\text{sp}^2$  carbon domains (G band) and  $\text{sp}^3$  (D band) hybridization inducing defects, respectively [17]. Compared with GO, the ratio of intensities ( $I_D/I_G$ ) for  $\text{Fe}_3\text{O}_4$ -graphene samples has a remarkable increase, and the G mode of  $\text{Fe}_3\text{O}_4$ -graphene becomes weaker and broader, suggesting a higher level of disorder of the graphene layers due to deoxygenation and introduction of  $\text{Fe}_3\text{O}_4$  during high temperature thermolysis [18]. Fig. 4(b) shows the XPS patterns of the GO and  $\text{Fe}_3\text{O}_4$ -graphene. The peaks  $\text{Fe}2p_{3/2}$  and  $\text{Fe}2p_{1/2}$  are located at  $\sim 712$  and

$\sim 724\text{ eV}$ , respectively, and not at  $\sim 710\text{ eV}$  which is for  $\gamma\text{-Fe}_2\text{O}_3$  [19]. This indicates the formation of the  $\text{Fe}_3\text{O}_4$  phase in the graphene matrix [20]. In addition, a decrease of O/C ratio with calcining time indicates a considerable reduction of GO.

The  $\text{Fe}_3\text{O}_4$ -graphene composite combines the high adsorption capacity of graphene and the separation convenience of magnetic nanoparticles, which makes it an excellent candidate for the removal of contaminants from water. As determined in Fig. 5(a), the BET specific surface area of  $\text{Fe}_3\text{O}_4$ -graphene composite is  $41\text{ m}^2/\text{g}$ . The relatively large specific surface area is beneficial for effective absorbent for removing contaminants in water. As shown in the top left inset of Fig. 5(b), after dispersing of  $\text{Fe}_3\text{O}_4$ -graphene composites in the MB solution and shaking the resultant mixture by hand for only 1 minute under natural lighting, the magnetic graphene nanosheets could be easily separated from the mixture by an external magnetic field and the solution became colorless. It can be considered that high sensitivity to the external magnetic field could be remarkably enhanced by decorating graphene sheets with magnetic  $\text{Fe}_3\text{O}_4$  nanoparticles [21]. The UV spectra show nearly complete MB removal from water (Fig. 4(b)).

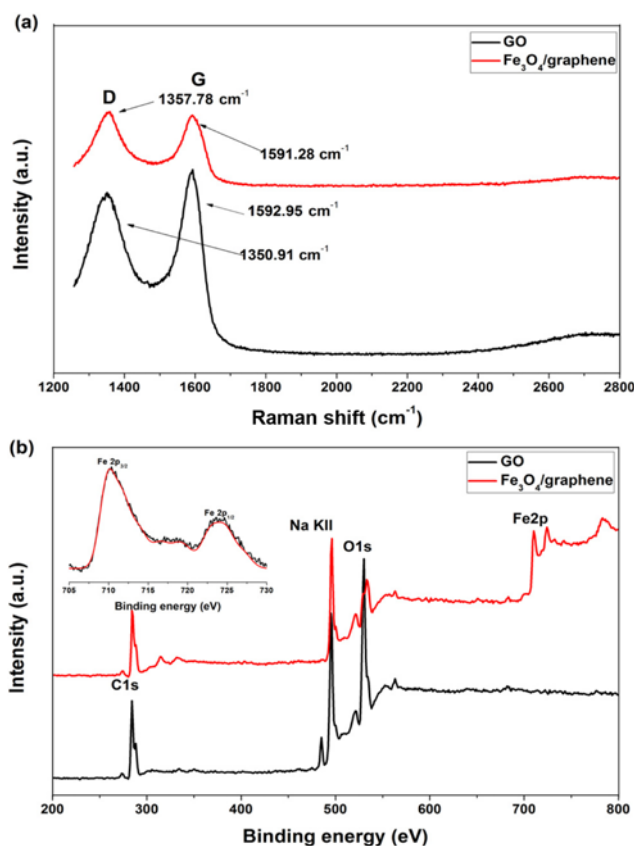


Fig. 4. Raman (a) and XPS (b) spectra of graphene and Fe<sub>3</sub>O<sub>4</sub>-graphene.

The advantages of Fe<sub>3</sub>O<sub>4</sub>-graphene composite for removal of MB from solution are investigated by comparing pure graphene and Fe<sub>3</sub>O<sub>4</sub>, as shown in Fig. 5. Graphene sheets were removed from the MB solution by centrifugation. Fe<sub>3</sub>O<sub>4</sub>-graphene and Fe<sub>3</sub>O<sub>4</sub> were removed from the MB solution by an external magnetic field. The data for adsorption of MB dye by different samples were measured at the same time intervals. The results show that the Fe<sub>3</sub>O<sub>4</sub>-graphene composite has fast adsorption kinetics towards MB dye solution within 20 min.

Finally, according to most of the synthetic approaches of mag-

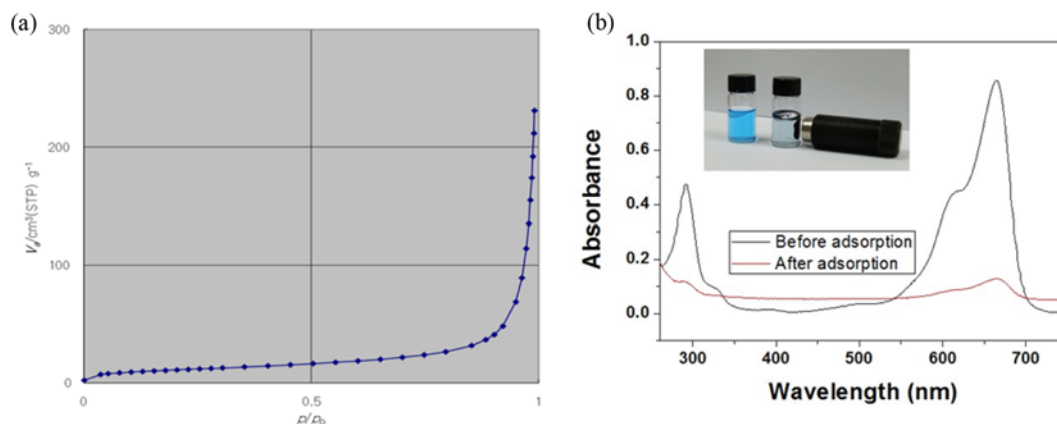


Fig. 5. N<sub>2</sub> adsorption isotherms of Fe<sub>3</sub>O<sub>4</sub>-graphene composite (a). UV-Vis spectra of MB solution before and after adsorption by Fe<sub>3</sub>O<sub>4</sub>-graphene (b). The inset in Fig. 4(b) shows the adsorption process of Fe<sub>3</sub>O<sub>4</sub>-graphene for MB.

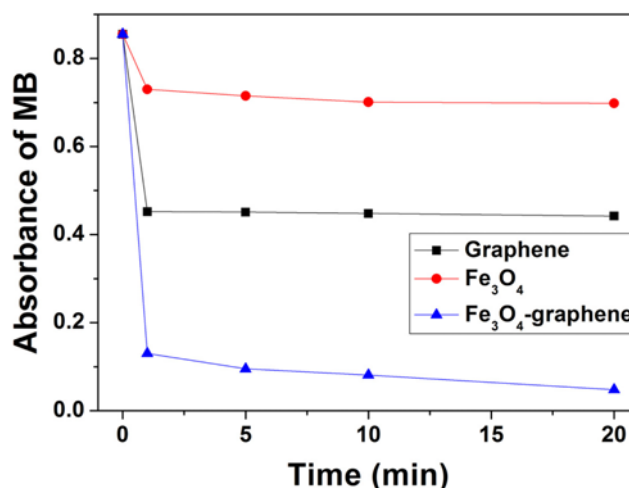


Fig. 6. Kinetic adsorption data plots for MB.

netic graphene documented so far, the thermolysis reaction mechanism for producing this material is still unclear. Despite many previously reported analogous cases in which magnetic graphene has been completed using organic compound by a hydrothermal method, the resulting composite usually has a mixed ( $\gamma$ -Fe<sub>2</sub>O<sub>3</sub>+Fe<sub>3</sub>O<sub>4</sub>) phase [3,22]. Although other researchers have used organic salt as the reaction initiator for synthesizing high crystalline Fe<sub>3</sub>O<sub>4</sub>-graphene, the reaction process still takes a long time [5,23,24]. The proposed mechanism for the synthesis is tentatively explained as follows: 1) the ferrocene can be attached onto GO surface through physical adsorption, electrostatic binding or charge transfer [25]. 2) At high temperature, GO is deoxygenated [26], the product being mainly CO<sub>2</sub> and H<sub>2</sub>O. 3) Because ferrocene is well soluble in Sc-CO<sub>2</sub> [27] at a temperature higher than that of thermolysis, the low valence iron cations in ferrocene were slowly oxidized by CO<sub>2</sub>, substituting the cyclopentadienyl-group and oxidizing Fe<sup>2+</sup> to high crystalline Fe<sub>3</sub>O<sub>4</sub>.

## CONCLUSION

We successfully synthesized high crystalline Fe<sub>3</sub>O<sub>4</sub>-graphene composites *via* a time-dependent thermolysis reaction, where high crystalline Fe<sub>3</sub>O<sub>4</sub> nanoparticles are hybridized with graphene nanosheets. With

the development of the preparation technology of Fe<sub>3</sub>O<sub>4</sub>-graphene, the simple procedure not only contributed to the cost of Fe<sub>3</sub>O<sub>4</sub>-graphene, but also provided high crystalline of Fe<sub>3</sub>O<sub>4</sub>-graphene for more applications in the near future. This paper is relevant and important for the application of a magnetic separation method and graphene in environmental pollution cleanup.

### ACKNOWLEDGEMENT

The authors thank for financial supports from the Doctor Foundation (2011), Open Foundation of Building Energy-saving Institute (Grant No. 2012-08) and College Students' Science and Technology Innovation Foundation of Anhui University of Architecture (2011).

### REFERENCES

1. H. Jabeen, V. Chandra, S. H. Jung, J. W. Lee, K. S. Kim and S. B. Kim, *Nanoscale*, **3**, 3583 (2011).
2. K. Zhang, V. Dwivedi, C. Y. Chi and J. S. Wu, *J. Hazard. Mater.*, **182**, 162 (2010).
3. J. F. Shen, Y. Z. Hu, M. Shi, N. Li, H. W. Ma and M. X. Ye, *J. Phys. Chem. C*, **114**, 1498 (2010).
4. H. P. Cong, J. J. He, Y. Lu and S. H. Yu, *Small*, **6**, 169 (2010).
5. X. P. Shen, J. L. Wu, S. Bai and H. Zhou, *J. Alloys. Compd.*, **506**, 136 (2010).
6. X. Y. Zhang, X. Y. Yang, Y. F. Ma, Y. Huang and Y. S. Chen, *J. Nanosci. Nanotechnol.*, **10**, 2984 (2010).
7. G. M. Zhou, D. W. Wang, F. Li, L. L. Zhang, N. Li, Z. S. Wu, L. Wen, G. Q. Lu and H. M. Cheng, *Chem. Mater.*, **22**, 5306 (2010).
8. X. Y. Yang, X. Y. Zhang, Y. F. Ma, Y. Huang, Y. S. Wang and Y. S. Chen, *J. Mater. Chem.*, **19**, 2710 (2009).
9. F. He, J. T. Fan, D. Ma, L. M. Zhang, C. Leung and H. L. Chan, *Carbon*, **48**, 3139 (2010).
10. V. Chandra, J. S. Park, Y. Chun, J. W. Lee, I. C. Hwang and K. S. Kim, *ACS Nano*, **4**, 3979 (2010).
11. V. Chandra and K. S. Kim, *Chem. Commun.*, **47**, 3942 (2011).
12. X. J. Deng, L. L. Lv, H. W. Li and F. Luo, *J. Hazard. Mater.*, **183**, 923 (2010).
13. H. M. Sun, L. Y. Cao and L. H. Lu, *Nano. Res.*, **4**, 550 (2011).
14. W. S. Hummers and R. E. Offeman, *J. Am. Chem. Soc.*, **80**, 1339 (1958).
15. S. H. Lee, H. W. Kim, J. O. Hwang, W. J. Lee, J. Kwon, C. W. Bielawski, R. S. Ruoff and S. O. Kim, *Angew. Chem. Int. Ed.*, **49**, 10084 (2010).
16. H. M. Ju, S. H. Choi and S. H. Huh, *J. Kor. Phys. Soc.*, **57**, 1649 (2010).
17. F. Tuinstra and J. L. Koenig, *J. Chem. Phys.*, **53**, 1126 (1970).
18. A. C. Ferrari and J. Robertson, *Phys. Rev. B*, **61**, 14095 (2000).
19. J. Lu, X. Jiao, D. Chen and W. Li, *J. Phys. Chem. C*, **113**, 4012 (2009).
20. D. Zhang, Z. Liu, S. Han, C. Li, B. Lei, M. P. Stewart, J. M. Tour and C. Zhou, *Nano. Lett.*, **4**, 2151 (2004).
21. J. E. Kim, T. H. Han, S. H. Lee, J. Y. Kim, C. W. Ahn, J. M. Yun and S. O. Kim, *Angew. Chem. Int. Ed.*, **50**, 3043 (2011).
22. J. F. Shen, M. Shi, H. W. Ma, B. Yan, N. Li and M. X. Ye, *Mater. Res. Bull.*, **46**, 2077 (2011).
23. X. Y. Li, X. L. Huang, D. P. Liu, X. Wang, S. Y. Song, L. Zhou, H. J. Zhang, *J. Phys. Chem. C*, **115**, 21567 (2011).
24. Y. Zhang, B. Chen, L. M. Zhang, J. Huang, F. H. Chen, Z. P. Yang, J. L. Yao and Z. J. Zhang, *Nanoscale*, **3**, 1446 (2011).
25. G. Williams, B. Seger and P. V. Kamat, *ACS Nano*, **2**, 1487 (2008).
26. D. X. Yang, A. Velamakanni, G. Bozoklu, S. J. Park, M. Stoller, R. D. Piner, S. Stankovich, I. H. Jung, D. A. Field, C. A. Ventrice Jr. and R. S. Ruoff, *Carbon*, **47**, 145 (2009).
27. C. M. Cowey, K. D. Bartle, M. D. Burford, A. A. Clifford and S. Zhu, *J. Chem. Eng. Data*, **406**, 1217 (1995).